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Self-branched α–MnO_2/δ–MnO_2 heterojunction nanowires with enhanced pseudocapacitance

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Keywords: α-MnO_2, δ-MnO_2, Self-branched structure, Supercapacitor, Heterojunction nanowires

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Abstract

Despite the extensive research on MnO_2 as the pseudocapacitor electrode material, there has been no report on heterostructures of multiple phase MnO_2. Here we report the combination of two high-capacitance phases of MnO_2, namely, α–MnO_2 nanowire and δ–MnO_2 ultrathin nanoflakes, to form a core-branch heterostructure nanoarray. This material and structure design not only increases the mass loading of active materials (from 1.86 to 3.37 mg cm^2), but also has resulted in evidently pseudocapacitance enhancement (from 28 F g^{-1} for pure nanowire to 178 F g^{-1} for heterostructures at 5 mV s^{-1}). The areal capacitance is up to 783 mF cm^{-2} at 1 mV s^{-1}. Upon 20,000 cycles, the heterostructures array electrode still deliver a reversible capacitance above 100 F g^{-1} at 4.5 A g^{-1}. Kinetic analysis reveals that capacitances due to both capacitive and diffusion controlled processes have been enlarged for the self-branched heterostructures array. This work paves a new way to push the electrochemical performance of MnO_2 as binder-free supercapacitor electrodes.

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Conceptual Insight

The performance of electrochemical energy storage devices relies largely on a scrupulous design of nanoarchitectures and smart hybridization of active materials. Nanoarray electrodes are particularly investigated for power source in microelectronics, which requires high rates, high areal capacity/capacitance and long cycle stability. MnO_2 has been the paradigm pseudo-capacitive material that is promising for practical applications. It has many crystal phases, among which α-MnO_2 and δ-MnO_2 have relatively high capacitance. Numerous studies have shown that crystal phase is an important factor to the electrochemical property of an electrode material. We constructed a bi-phase heterostructure nanoarray electrode by growing thin nanoflakes of δ-MnO_2 to α-MnO_2 nanowire core. This not only increases the mass loading to the highest value compared to literature, but also significantly increased the areal capacitance. Interestingly, the capacitance originates from both the well-known pseudocapacitive process and Li-ion diffusion process due to the large lattice spacing of the δ-phase nanosheets.

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Introduction

There are tremendous demands for energy storage devices that exhibit both high energy density and meanwhile greater power density than the state-of-art lithium ion batteries.[1-4] As the typical high power candidate, supercapacitors can also deliver energy density comparable to batteries by utilizing high-capacitance electrode materials according to the equation $E = \frac{1}{2}CV^2$, where E is energy stored in the device, C is capacitance and V is the open-circuit potential.[5-10] For supercapacitors, the aim is to boost the energy density without sacrificing the high power capability. In this regard, transition metal oxides with generally high theoretical capacitances are of high pertinence. Manganese dioxide (MnO_2) is one of typical pseudo-capacitive materials. [11-14] In an aqueous electrolyte, the redox reaction between MnO_2 and cations follows $MnO_2 + M^+ + e^- \Leftrightarrow MnOOM$ (M^+ stands for H_3O^+ or alkali cations such as K^+, Na^+ and Li^+),[15-18] which delivers a theoretical psedocapacitance of ~1370 F g^{-1}. Furthermore, because of the multi-modality of the octahedral [MnO_6] basic unit linkage, MnO_2 forms a variety of crystal phases (α, β, γ and δ etc.).[11, 12, 14, 19-27] Specifically, 3D tunneling structures are formed in α (2×2 and 1×1),[20, 21, 23] β (1×1)[19] and γ (2×1)[12, 27] phases whereas the δ-MnO_2 possesses a 2D layered structure.[14, 22, 24-26] Accordingly, the tunnel size in α-MnO_2 and δ-MnO_2 are 0.46 nm and 0.7 nm, respectively, both much larger than that in β-MnO_2 (0.23 nm).[11, 12, 19, 20] Previous study has shown that α-MnO_2 and δ-MnO_2 are of higher capacitance than the β and γ counterparts.[28] Hence, utilizing the α-MnO_2 and δ-MnO_2 as supercapacitor electrodes should be beneficial to achieving high capacitance.

So far there is a vast literature of supercapacitive MnO_2 in which capacitance values close to the theoretical quantity of 1370 F g^{-1} have been reported.[13, 14, 21, 23, 25, 29-34] However, most of these studies utilized a rather thin MnO_2 film with mass loadings less than 0.5 mg cm^{-2}. Therefore, the total capacitance (in unit of F) and amount of charges (in unit of C) are small, which will not meet certain applications that require high areal and/or volumetric capacitance. On the other hand, active materials directly grown on current collectors are regarded as important to the construction of high energy density electrodes. Binders (such as PVDF) and conductive agents (such as acetylene black) lower the overall capacitance of the entire electrode because they reduce the ratio of active materials. So far, growing of

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single-phase MnO_2 nanostructures directly on conductive substrate such as graphene, graphene foam, carbon nanotube and carbon cloth have been reported by electrodeposition or hydrothermal methods.[26, 27, 34-38] These array electrodes are all based on single-phase MnO_2 and the carbon materials do not contribute to the capacitance. There is no report yet on multiple phase heterostructures for high-capacitance electrodes.

A previous study demonstrates a synergistic effect between anatase and TiO_2-B in bi-phase TiO_2 nanosheets which accounts for the capacity increase for lithium ion storage.[39] Inspired by this, we intend to enhance the capacitance of MnO_2 by optimizing the phase combination in their array structure. Our mature synthesis capability enables us to construct core-branch nanostructures with a mass loading up to 3.37 mg cm^{-2}. In the self-branched heterostructures, the intrinsic merits of large capacitance of both phases can be more efficiently utilized. Based on above design, the core-branch electrode indeed shows much higher specific capacitance at all different scan rates than single-phase nanowires. In addition, it exhibits high capacitance retention (~86 %) after a 20000 cycles @ 4.5 A g^{-1}. These properties (cycling stability and capacitance) are considered superior among all the binder-free MnO_2 capacitive electrodes.[36-38]

Experimental

Material preparation

Carbon cloth containing highly ordered weaved carbon fiber with diameter ~5 μm was used as current collector and conductive substrate to support the electrode active material. The carbon cloth was purchased from Cetech Tai Wan. The thickness as provided by the manufacturer is ~ 300 μm. The carbon cloth was treated in air plasma (room temperature, 230 psi for 10 min) to make the surface hydrophilic. Then MnO_2 nanowires were grown through a hydrothermal synthesis. In details, 0.25 g $KMnO_4$, 0.048 g $Fe_2(SO_4)_3$ and 18 μl concentrated H_2SO_4 were mixed in 80 ml distilled water with magnetic stirring under room temperature for 20 min. The resulted solution was transferred into 50 ml autoclave, which was kept in a box furnace under 150 ^oC for 6 h. Then the obtained materials were thoroughly washed with distilled water and dried in an electrical oven (80 ^oC, 8h). Afterwards, in order to grow the

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MnO_2 branch, the samples underwent a second hydrothermal process. The procedure is similar as above except for the solution (herein pure $KMnO_4$ 0.2844 g) and the reaction condition (160 ^oC, 1 h). Finally, both the as-synthesized MnO_2 nanowires and MnO_2 core-branch sample were annealed in Ar at 250 ^oC for 2 h.

Structural and morphology characterization

The crystal structures of the samples were identified using XRD (RigakuD/Max-2550 with Cu Kα radiation) and Raman (Renishaw InVia Raman microscope). Raman spectra were recorded using a 633 nm laser excitation source and a ×50 objective lens. The exposure time was limited to 30s with 5 accumulations to minimize the beam damage. Different spots across the sample were examined to confirm the consistency of the results. Fits of Raman spectra were performed by PeakFit software (version 4.12) automatically with the assumptions of linear background and all Raman bands have a Lorentz line shape. The morphology of all the samples was examined by an Ultra-High Resolution Scanning Electron Microscopy (FEI XL30 SFEG UHR SEM) at 10 kV and TEM/HRTEM (JEOL JEM-2100F) operating at 200 kV. AFM measurements were conducted with a Veeco Scanning Probe Microscope in tapping mode. The surface areas of the samples were determined by gas adsorption analyses using Quantachrome Instruments Autosorb-iQ (Boynton Beach, Florida USA, gas purity ~ 99.99%, samples were outgassed at 150 ^oC for 6 h for activation). The specific surface area was calculated with the Brunauer–Emmett–Teller (BET) model. The pore size distribution was determined using the Barrett–Joyner–Halenda (BJH) method.

Electrochemical test and mass determination

The electrochemical tests were conducted with a standard three-electrode testing system where the electrode materials (1×1 cm^2) acted as working electrode, Pt as counter electrode and Ag/AgCl as reference electrode in 1 M LiCl electrolyte (Salartron). The active material mass loading was determined by weighing the samples after each step of the synthesis procedure using a high-precision balance (0.001 mg). The mass loading of the self-branched MnO_2 electrode on the carbon cloth is about 3.37 mg cm^{-2} and about 1.86 mg cm^{-2} for the pure MnO_2 nanowire sample.

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Calculation of specific capacitance

The specific gravimetric capacitances (C_m) are calculated based on the CV curves according to the following equations:

$$C_m = \frac{Q}{\Delta U \times m} = \frac{\int I_m dU}{2v\Delta U}$$

where C_m (F g^{-1}) is the specific capacitance, Q (C) is the average charge, ΔU (V) is the potential window, v (V s^{-1}) is the scan rate, and I_m (A g^{-1}) is the gravimetric current of the working electrode (i.e. the vertical coordinate of the CV curve). There is a factor 2 because one CV cycle includes charge and discharge process, and the integral contains the charge of these two processes.

Results and discussion

Figure 1 illustrates the synthesis process of the resulted self-branched MnO_2 (SBM) electrode with 3D tunneled structure in its core and 2D layered structure in the branch, forming by different stacking of the [MnO_6] octahedral basic unit. In the first hydrothermal process, alpha-phase nanowire MnO_2 (NWM) is designed to grow directly on a piece of conductive carbon cloth substrate and acts as the backbone (**Figure 1a**). Then, the entire layout is completed with delta-phased MnO_2 nanoflake fully anchored to the MnO_2 nanowire skeleton (**Figure 1b**).

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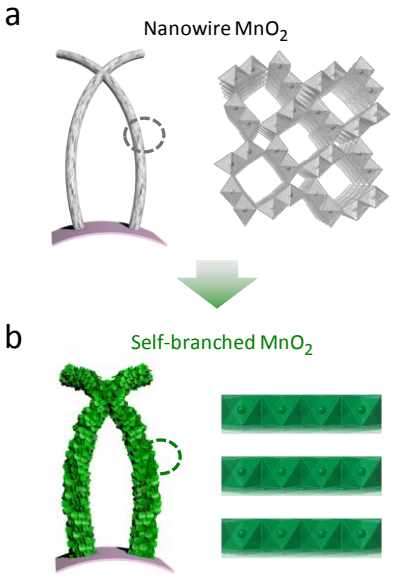

Figure 1. Illustration of electrode design. a. Alpha-MnO_2 (with 2×2 and 1×1 channels) nanowires on carbon cloth; b. Self-branched MnO_2 electrode formed after delta-MnO_2 (with layered structure) flakes were decorated to the alpha-MnO_2 nanowire.

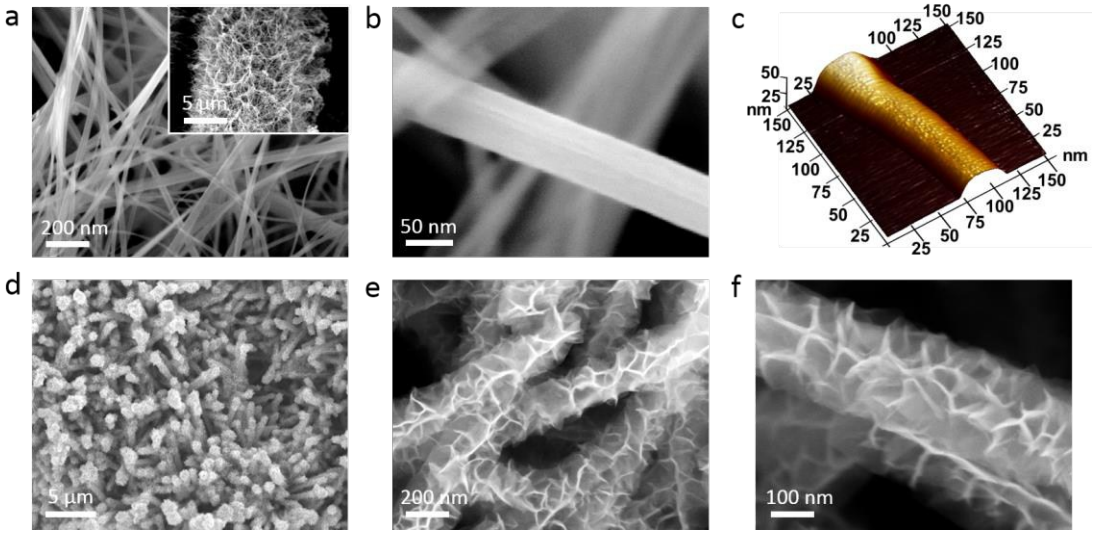

Figure 2. Morphology. SEM images of a-b) NW MnO_2. Inset of a is the nanowire under lower magnification showing the homogeneous distribution of the wire on carbon fiber. c) AFM 3D image of a single nanowire. d-e) As-synthesized SB MnO_2 electrode and f) after annealing.

We unveil the morphology of electrode materials under scanning electron microscope (SEM). After plasma treatment in air for 10 min, the carbon cloth surface becomes hydrophilic. This is a necessity step for the nucleation and growth of MnO_2 in this study (as

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well as of course other active materials in hydrothermal growth). The first-stage hydrothermal synthesis results in MnO_2 nanowire arrays that grow homogeneously on the carbon cloth (**Figure 2a**). The length of the nanowire is ~ 5 μm, which to our best knowledge is the longest among all the reported 1D MnO_2 fabricated through solution based methods.[23, 40] A possible reason for this different sample morphology is that they were grown under a highly concentrated heating condition of the box furnace whereas traditional hydrothermal growth is usually conducted in an electric oven. Higher magnification SEM images (**Figure 2b**), together with the atomic force microscope (**Figure 2c** and **Figure S1**), further confirm the solid nanowire morphology. After the second hydrothermal procedure, the nanowire skeleton is well reserved (**Figure 2d**) and uniformly decorated with thin flakes, forming a nanoflake @ nanowire core-branch structure (**Figure 2e**). To tighten the connection of the nanoarrays to the carbon cloth substrate, the sample after hydrothermal growth was further annealed in an Ar atmosphere at 250 ^oC for 2 h. As seen from **Figure 2f**, the final core-branch array structure is retained.

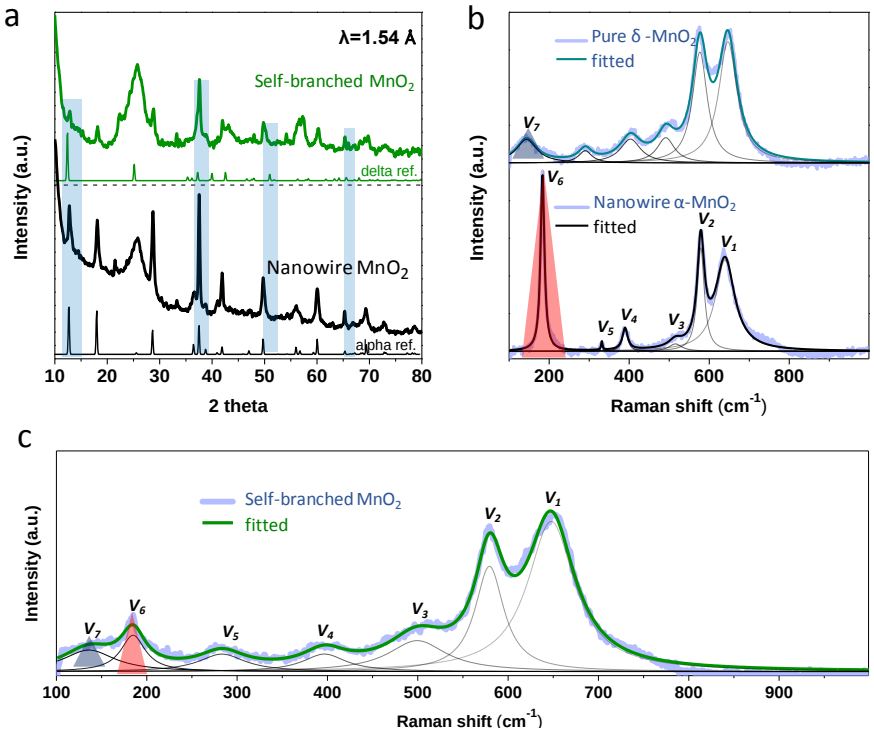

Figure 3. Crystal structure. a) XRD spectra of NWM and SBM electrodes. The vertical bars indicate the common peaks appearing in both samples. Raman spectra of b) the alpha (nanowire), delta phase

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(controlled sample) MnO_2 and c) SBM electrode. The fitted peak positions are listed in the table S1 in supporting information.

In order to confirm the two phases in the core-branch electrode, we first look at X-ray diffraction (XRD) result. Because of the similar interlayer spacing values of MnO_2 alpha and delta phases (as shown by the calculated spectra), many of the diffraction peaks appear at the same position, as seen in **Figure 3a**. For example, α-MnO_2 (110) and δ-MnO_2 (001) locate at $2\theta = 12.6^o$; α-MnO_2 (400) and δ-MnO_2 (110) locate at $2\theta = 36.3^o$, α-MnO_2 (411) and δ-MnO_2 (-112) locate at $2\theta = 50.3^o$, and α-MnO_2 (002) and δ-MnO_2 (-312) locate at $2\theta = 65.3^o$ (α-MnO_2 ref. JCPDS #44-0141; δ-MnO_2 ref. JCPDS #42-1317).[14, 20-26, 30] Therefore, the key to differentiate these two phases is the δ-MnO_2 (002) peak located at $2\theta = 25.4^o$. Unfortunately, this peak overlaps with the main peak of carbon from the substrate.

We seek further verification with Raman spectroscopy, in the consideration that manganese dioxide is very Raman active because of its short-range ordering crystalline structure. According to the group theoretical calculation, alpha-MnO_2 (I4/m space group) has fifteen vibration modes ($6A_g+6B_g+3E_g$) while delta-MnO_2 (C2/m space group) has nine Raman active modes ($3A_g+6B_g$). Because of the low polarizabilities of some vibration modes and the overlapping of incompletely resolved modes, we have to differentiate alpha and delta phase MnO_2 at their low frequency regions.[41-43] Based on previous research, we fit the Raman spectra of α- and δ- manganese dioxide with Lorentz fitting which locates seven strong and weak Raman bands basically.[44, 45] The fitted results (**Figure 3b**) point out the difference between the α- and δ- MnO_2 clearly because α-MnO_2 monopolizes the V_6 (183 cm^{-1}) while δ-MnO_2 owns the broad peak V_7 (130 ~150 cm^{-1}). Because of the high similarities, α- and δ-MnO_2 share the Raman peaks from V_1 to V_5 (see the peak positions listed in table S1 and detailed Raman fitting analysis in supporting information). All the peaks are detected in SBM (**Figure 3c**) including the key different near region peaks of V_6 and V_7. Thus the SBM can be confirmed containing both α- and δ- MnO_2.

To provide more evidence of the morphology and crystal structure, transmission electron microscopy (TEM) characterization was conducted for both MnO_2 electrode materials. The low magnification TEM images confirmed the nanowire (**Figure 4a**) and nanowire @

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nanoflake core-shell (**Figure 4c**) structure of the NWM and SBM, respectively. Regarding the pure MnO_2 nanowire, the high resolution TEM characterization exhibits the (110) lattice of α-MnO_2 (d = 6.9 Å) in **Figure 4b**, corresponding well with the XRD data in **Figure 3a**. Furtherly, the fast Fourier transfer (FFT) image generated from the same area demonstrates its single crystalline property (**Figure 4b inset**). Moreover, in the HRTEM images of the SBM sample, typical planes of δ-MnO_2-(110) (d = 2.5 Å), δ-MnO_2-(200) (d = 2.6 Å) and δ-MnO_2-(-114) lattice (d = 1.5 Å) are identified on the edge region (Figure 4d), confirming the existence of δ-MnO_2 in the SBM electrode. Meanwhile, the TEM element mapping show the homogeneous distributions of Mn and O match well with the dark-field image of the SBM profile (**Figure 4e**). Hence, the co-existence of two MnO_2 phases in the SBM structure is further manifested. Therefore, both the α-MnO_2 @ δ-MnO_2 core-branch and the pure α-MnO_2 nanowire structures are verified.

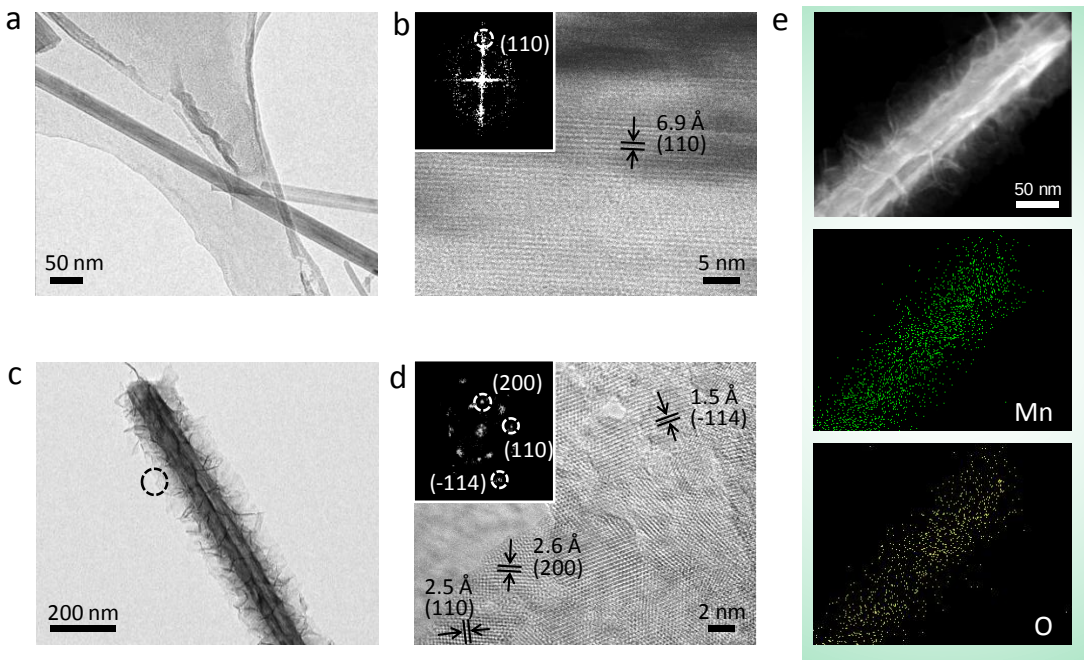

Figure 4. Morphology and crystal information study with TEM. TEM images of a) NWM and c) SBM nanostructure; HRTEM images of b) NWM and d) SBM nanostructure with insets the fast Fourier transfer (FFT) generated from the same area; e) TEM element mapping of a single SBM wire.

Nanoflakes grown on the nanowire can increase the active material mass loading and generally ensure an improved surface area of the electrode. To confirm this, gas adsorption

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method was employed to characterize the specific surface area (**Figure S2**). Under the Brunauer–Emmett–Teller (BET) model, the surface area of the SBM nanostructure is calculated to be 60.7 m^2 g^{-1}, which is six times that of NWM (10.2 m^2 g^{-1}). The 2 × 2 tunnels (0.46 nm) in α-MnO_2 and the layered structure (0.69 nm) of δ-MnO_2 are both beneficial in facilitating ion insertion and transport. Therefore, we foresee that the combination of these two phases can secure an improved rate performance and an increased capacitance.

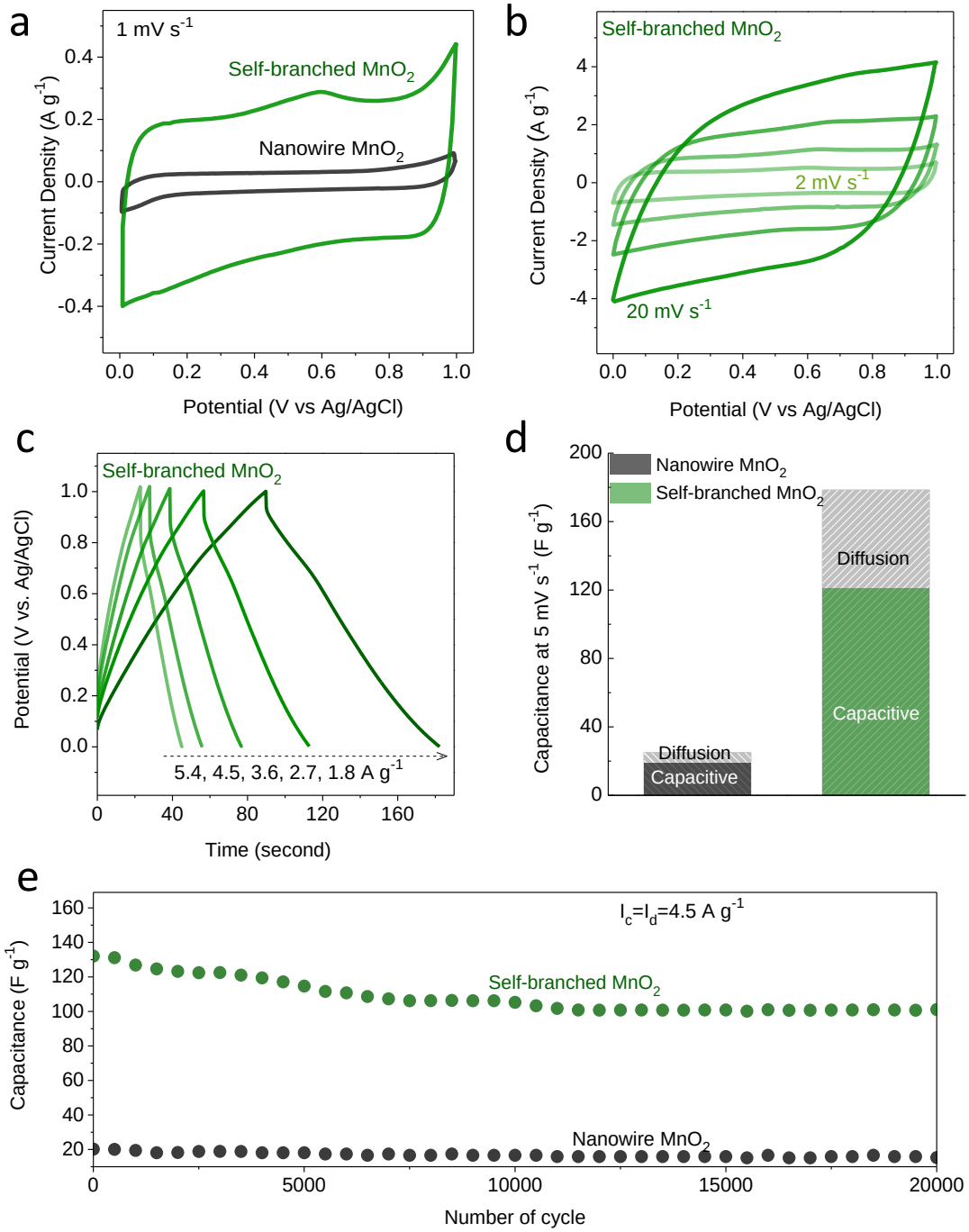

Figure 5. Electrochemical properties. a) CV curves comparison of NWM and SBM nanostructure at 1

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mV s^{-1}; b) CV curves of SBM nanostructure at different scan speeds; c) Charge-discharge curves of SBM nanostructure at varied current density; d) capacitance contribution calculation of MnO_2 electrodes at 5 mV s^{-1}. e) cycling performance of NWM and SBM nanostructure in 20,000 cycles under 4.5 A g^{-1}.

To evaluate the structure design, the electrochemical properties of the pure α-MnO_2 nanowires and bi-phase core-branch electrodes as supercapacitive electrodes are tested for comparison. Both electrodes are of quasi-rectangular-shaped capacitive cyclic voltammetry (CV) curves (**Figure 5a**) at 1 mV s^{-1}. The capacitance for the pure MnO_2 nanowire is rather small, as indicated by the low current density value of the CV curves (**Figure S3a**). In contrast, after the decoration with MnO_2 nanoflakes, the specific capacitances are drastically increased as seen from the comparison in **Figure 5a**. Moreover, when the scan speed increases from 2 to 20 mV s^{-1}, the SBM still exhibits quasi-rectangular shaped CV curves (**Figure 5b**), manifesting the dominating capacitive behavior. It is worth mentioning that the capacitance increase of the SBM electrode compared to the pure nanowire electrode occurs also consistently at all different scan speeds (from 1, 2, 5, 10, to 20 mV s^{-1}). Their corresponding values obtained from the CV curves are (SBM vs α-MnO_2 nanowire): 232 vs 32 F g^{-1}, 208 vs 28 F g^{-1}, 179 vs 25 F g^{-1}, 152 vs 23 F g^{-1} and 116 vs 22 F g^{-1} (Figure 5b and S3a). Galvanostatic charge-discharge curves of both electrodes are in triangular shape (**Figure 5c** and **Figure S3b**), further verifying the dominating capacitive behavior. Coulombic efficiency (C.E.) calculated from charge/discharge curves (**Figure S5)** shows that the C.E. of both SBM and NWM electrodes are above 95% under various current densities. The obvious capacitance loss at 1 A g^{-1} could be due to the irreversible Li^+ insertion. At rates higher than 3 A g^{-1}, the insignificant capacitance loss is proposed to be related to oxygen evolution reaction. Upon prolonged cycling up to 20,000 cycles, the core-branch nanowire electrode shows nearly overlapping charge and discharge capacitances with reasonable stability (retention ~86 %, see **Figure 5e**). This retention is somewhat inferior to that of pure MnO_2 nanowire electrode (~98 %), but the capacity values are all the way significantly higher (e.g., 100 F g^{-1} versus 20 F g^{-1}).

To better understand the capacitance enhancement effect of combining these two phases, δ-MnO_2 nanoflakes were also synthesized on carbon paper for comparison (data are presented in **Figure S4**). The specific capacitances calculated from the CV curves were

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plotted together with the pure alpha MnO_2 nanowire and the SBM electrode. It is clear that the SBM composite electrode exhibits higher capacitance than both MnO_2 nanowire and nanoflake in different scan speeds (**Figure S4b**), demonstrating the advantage of combining these two phases into a core-branch architecture. We ascribe this capacitance enhancement essentially to the increased surface area and also a synergistic effect. The alpha and delta phase MnO_2 may have a poor-defined crystallographic interface as they may diffuse into the lattice of each other and cause lattice defects, which is common for heterostructures. The defects will lead to further exposure of electrochemical active sites and to increased capacitance, similar to previous studies.[39, 46, 47]

For powder-packed electrodes, an increase in material loading will generally lead to a reduction in the specific capacitance by lowering the electrode utilization efficiency. On the other hand, nanoscale electrode materials exhibit both diffusion-controlled current and pure capacitive one; the enlarged surface area and shortened diffusion distance may lead to fast kinetics thus contribute to capacitive current. In our case, a marriage of the nanoflake and nanowire forming 3D hierarchical array structure does not eliminate effective interfaces between electrode and electrolytes, instead, it would provide a fast ion transport for the surface redox reactions. This will facilitate ion diffusion and lead to higher capacitive current than the pure nanowire electrode. To verify this, we calculated the capacitances based on the CV curves obtained from various scan speeds following the standard method.[48-56] One example at 5 mV s^{-1} is selected for analysis (**Figure 5d and Figure S6**). The capacitance contribution in the SBM electrode from capacitive behavior is nearly ten times that of the NWM electrode (~ 120 vs 19 F g^{-1}). Similar results have been obtained for other scan speeds, i.e., ~8 times increase in capacitive process induced capacitance. In addition, we also notice the enhancement in capacitance due to the battery insertion process, in which Li ions diffuse swiftly and reversibly into lattice channels. For example, in Figure 5d the "diffusion contributed capacitance" increases from 9 to 58 F g^{-1}. This additional capacitance can also be ascribed to the δ-MnO_2 nanoflake branch, which has a well-defined diffusion channels for Li ions with a 0.69 nm lattice spacing. All in all, the above results demonstrate clearly that the judicious combination of the bi-phase MnO_2 nanocrystals have increased the entire mass loading of MnO_2 without sacrificing the capacitance; Instead, the total capacitance due to

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both capacitive and diffusion processes has been enlarged.

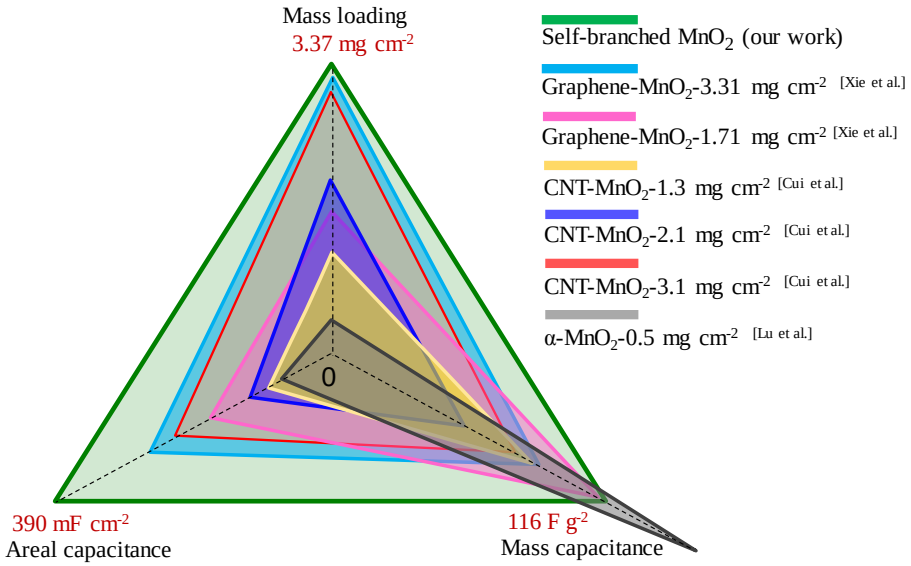

Figure 6. Capacitance comparison with previous studies considering mass loading. All the capacitance values were obtained at a scan rate of 20 mV s^{-1}.

To benchmark our result, the stereogram of gravimetric capacitance and areal capacitance against mass loading is plotted in **Figure 6** in comparison to previous work on MnO_2 nanostructure array electrodes (their values are also tabulated in Table S2, supporting information).[13, 23, 37, 57] The comparison shows that our SBM electrode has the highest areal capacitance and also relatively high gravimetric capacitance despite the highest mass loading. This is very encouraging as MnO_2 is one of promising materials for supercapacitor applications. It is optimistic that such bi-phase MnO_2 synthesized into other nanostructures, such as core-shell hollow particles and particle-anchored nanoflakes, may also deliver superior performance in supercapacitive charge storage.

Conclusion

To sum up, for the first time we have realized bi-phase self-branched MnO_2 nanoarray structure, which is proven effective in boosting up the MnO_2 capacitance. The α-MnO_2 nanowire core and δ-MnO_2 nanoflake branch have different crystal structures and capacitive properties, and a positive synergistic effect has probably accounted for the drastic capacitance

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enhancement. The self-branched electrode presents fairly stable cycling with ~86 % capacitance retention in 20000 cycles. Through kinetic analysis of the charge storage behaviors, we have quantitatively identified that the self-branched electrode exhibit enlarged capacitances due to both capacitive and ion diffusion processes. We ascribe this enhancement to the self-branched configuration with δ-MnO_2 thin flakes. This material design paves the way for creating electrodes with high mass loading by utilizing multi-phased metal oxides heterostructures, including but not limited to TiO_2 and SnO_2.

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ACKNOWLEDGEMENTS

H. J. Fan acknowledges the financial supported by MOE AcRF Tier 1 (RG98/15 and RG115/15) and CAS/SAFEA International Partnership Program for Creative Research Teams. H.Z. thanks the support from MOE under AcRF Tier 2 (ARC 26/13, No. MOE2013-T2-1-034; ARC 19/15, No. MOE2014-T2-2-093; MOE2015-T2-2-057), and NTU under Start-Up Grant (M4081296.070.500000) in Singapore. UCSD team acknowledge the partial funding support by the Advanced Research Projects Agency – Energy (ARPA-E), U.S. Department of Energy, under Award Number DE-AR0000535. This work was performed in part at the San Diego Nanotechnology Infrastructure (SDNI), a member of the National Nanotechnology Coordinated Infrastructure, which is supported by the National Science Foundation (Grant ECCS-1542148).

Appendix A.

Supporting information Supplementary data associated with this article can be found in the on line version at http://dx.doi.org/10.1016/xxxx.

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Reference

1. V. Augustyn, P. Simon and B. Dunn, *Energy Environ. Sci.*, 2014, **7**, 1597-1614.
2. L. Liu, Z. Niu and J. Chen, *Chem. Soc. Rev.*, 2016, **45**, 4340-4363.
3. Y. Wang, Y. Song and Y. Xia, *Chem. Soc. Rev.*, 2016.
4. C. Zhu, P. Yang, D. Chao, W. Mai and H. J. Fan, *ChemNanoMat*, 2015, **1**, 458-476.
5. G. Wang, L. Zhang and J. Zhang, *Chem. Soc. Rev.*, 2012, **41**, 797-828.
6. J. Yan, Q. Wang, T. Wei and Z. Fan, *Adv. Energy Mater.*, 2014, **4**, 1300816.
7. A. S. Arico, P. Bruce, B. Scrosati, J.-M. Tarascon and W. van Schalkwijk, *Nat. Mater.*, 2005, **4**, 366-377.
8. B. G. Choi, M. Yang, W. H. Hong, J. W. Choi and Y. S. Huh, *ACS Nano*, 2012, **6**, 4020-4028.
9. S. Faraji and F. N. Ani, *J. Power Sources*, 2014, **263**, 338-360.
10. C. D. Lokhande, D. P. Dubal and O.-S. Joo, *Curr. Appl. Phys.*, 2011, **11**, 255-270.
11. S. Chen, J. Zhu, Q. Han, Z. Zheng, Y. Yang and X. Wang, *Cryst. Growth Des.*, 2009, **9**, 4356-4361.
12. S. Chou, F. Cheng and J. Chen, *J. Power Sources*, 2006, **162**, 727-734.
13. L. Hu, W. Chen, X. Xie, N. Liu, Y. Yang, H. Wu, Y. Yao, M. Pasta, H. N. Alshareef and Y. Cui, *ACS Nano*, 2011, **5**, 8904-8913.
14. J. Liu, J. Jiang, C. Cheng, H. Li, J. Zhang, H. Gong and H. J. Fan, *Adv. Mater.*, 2011, **23**, 2076-2081.
15. O. Ghodbane, J.-L. Pascal and F. Favier, *ACS Appl. Mater. Interfaces*, 2009, **1**, 1130-1139.
16. Y. Liu, X. Miao, J. Fang, X. Zhang, S. Chen, W. Li, W. Feng, Y. Chen, W. Wang and Y. Zhang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 5251-5260.
17. V. Subramanian, H. Zhu, R. Vajtai, P. Ajayan and B. Wei, *J. Phys. Chem. B*, 2005, **109**, 20207-20214.
18. J. Yan, Z. Fan, T. Wei, W. Qian, M. Zhang and F. Wei, *Carbon*, 2010, **48**, 3825-3833.
19. H. Jiang, T. Zhao, J. Ma, C. Yan and C. Li, *Chem. Commun.*, 2011, **47**, 1264-1266.
20. J. Liu, J. Essner and J. Li, *Chem. Mater.*, 2010, **22**, 5022-5030.
21. R. Liu, J. Duay and S. B. Lee, *ACS Nano*, 2010, **4**, 4299-4307.
22. Y. Liu, Y. Qiao, W. Zhang, H. Wang, K. Chen, H. Zhu, Z. Li and Y. Huang, *J. Mater. Chem. A*, 2015, **3**, 7780-7785.
23. W. Xiao, H. Xia, J. Y. H. Fuh and L. Lu, *J. Power Sources*, 2009, **193**, 935-938.
24. C. Xu, Y. Zhao, G. Yang, F. Li and H. Li, *Chem. Commun.*, 2009, 7575-7577.
25. W. Yan, T. Ayvazian, J. Kim, Y. Liu, K. C. Donavan, W. Xing, Y. Yang, J. C. Hemminger and R. M. Penner, *ACS Nano*, 2011, **5**, 8275-8287.
26. X. Zhang, X. Chang, N. Chen, K. Wang, L. Kang and Z.-h. Liu, *Journal of Materials Science*, 2012, **47**, 999-1003.
27. X. Zhao, L. Zhang, S. Murali, M. D. Stoller, Q. Zhang, Y. Zhu and R. S. Ruoff, *ACS Nano*, 2012, **6**, 5404-5412.
28. S. Devaraj and N. Munichandraiah, *J. Phys. Chem. C*, 2008, **112**, 4406-4417.
29. J. N. Broughton and M. J. Brett, *Electrochim. Acta*, 2004, **49**, 4439-4446.
30. Q. Fan and M. S. Whittingham, *Electrochem. Solid-State Lett.*, 2007, **10**, A48-A51.
31. J.-H. Kim, K. H. Lee, L. J. Overzet and G. S. Lee, *Nano Lett.*, 2011, **11**, 2611-2617.
32. H.-S. Nam, J. S. Kwon, K. M. Kim, J. M. Ko and J.-D. Kim, *Electrochim. Acta*, 2010, **55**, 7443-7446.
33. H. Xia, J. Feng, H. Wang, M. O. Lai and L. Lu, *J. Power Sources*, 2010, **195**, 4410-4413.
34. K. Zhang, X. Han, Z. Hu, X. Zhang, Z. Tao and J. Chen, *Chem. Soc. Rev.*, 2015, **44**, 699-728.
35. W.-y. Zou, W. Wang, B.-l. He, M.-l. Sun and Y.-s. Yin, *J. Power Sources*, 2010, **195**, 7489-7493.

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Materials Horizons Accepted Manuscript

View Article Online
DOI: 10.1039/C6MH00556J

36. Y. Cheng, S. Lu, H. Zhang, C. V. Varanasi and J. Liu, *Nano Lett.*, 2012, **12**, 4206-4211.
37. Y. He, W. Chen, X. Li, Z. Zhang, J. Fu, C. Zhao and E. Xie, *ACS Nano*, 2012, **7**, 174-182.
38. P. Lv, Y. Y. Feng, Y. Li and W. Feng, *J. Power Sources*, 2012, **220**, 160-168.
39. Q. Wu, J. Xu, X. Yang, F. Lu, S. He, J. Yang, H. J. Fan and M. Wu, *Adv. Energy Mater.*, 2015, **5**.
40. H. Pan, Y. Shao, P. Yan, Y. Cheng, K. S. Han, Z. Nie, C. Wang, J. Yang, X. Li, P. Bhattacharya, K. T. Mueller and J. Liu, *Nat. Energy*, 2016, 16039.
41. R. Baddour-Hadjean and J.-P. Pereira-Ramos, *Chem. Rev.*, 2009, **110**, 1278-1319.
42. T. Gao, M. Glerup, F. Krumeich, R. Nesper, H. Fjellvåg and P. Norby, *J. Phys. Chem. C*, 2008, **112**, 13134-13140.
43. C. Julien, M. Massot, R. Baddour-Hadjean, S. Franger, S. Bach and J. Pereira-Ramos, *Solid State Ionics*, 2003, **159**, 345-356.
44. D. Chen, D. Ding, X. Li, G. H. Waller, X. Xiong, M. A. El-Sayed and M. Liu, *Chem. Mater.*, 2015, **27**, 6608-6619.
45. T.-H. Wu, D. Hesp, V. Dhanak, C. Collins, F. Braga, L. J. Hardwick and C.-C. Hu, *J. Mater. Chem. A*, 2015, **3**, 12786-12795.
46. Y. Zheng, Y. Jiao, M. Jaroniec and S. Z. Qiao, *Angew. Chem. Int. Ed.*, 2015, **54**, 52-65.
47. D. Yang, H. Liu, Z. Zheng, Y. Yuan, J.-c. Zhao, E. R. Waclawik, X. Ke and H. Zhu, *J. Am. Chem. Soc.*, 2009, **131**, 17885-17893.
48. V. Augustyn, J. Come, M. A. Lowe, J. W. Kim, P.-L. Taberna, S. H. Tolbert, H. D. Abruña, P. Simon and B. Dunn, *Nat. Mater.*, 2013, **12**, 518-522.
49. R. Li and J. Liu, *Electrochim. Acta*, 2014, **120**, 52-56.
50. J. Yang, S. Muhammad, M. R. Jo, H. Kim, K. Song, D. A. Agyeman, Y.-I. Kim, W.-S. Yoon and Y.-M. Kang, *Chem. Soc. Rev.*, 2016.
51. J. W. Kim, V. Augustyn and B. Dunn, *Adv. Energy Mater.*, 2012, **2**, 141-148.
52. X. Wang, G. Li, Z. Chen, V. Augustyn, X. Ma, G. Wang, B. Dunn and Y. Lu, *Adv. Energy Mater.*, 2011, **1**, 1089-1093.
53. J. Come, V. Augustyn, J. W. Kim, P. Rozier, P.-L. Taberna, P. Gogotsi, J. W. Long, B. Dunn and P. Simon, *J. Electrochem. Soc.*, 2014, **161**, A718-A725.
54. C. Zhang, R. Maloney, M. R. Lukatskaya, M. Beidaghi, B. Dyatkin, E. Perre, D. Long, W. Qiao, B. Dunn and Y. Gogotsi, *J. Power Sources*, 2015, **274**, 121-129.
55. Z. Chen, V. Augustyn, X. Jia, Q. Xiao, B. Dunn and Y. Lu, *ACS Nano*, 2012, **6**, 4319-4327.
56. D. Chao, C. Zhu, P. Yang, X. Xia, J. Liu, J. Wang, X. Fan, S. V. Savilov, J. Lin, H. J. Fan and Z. X. Shen, *Nat. Commun.*, 2016, **7**, 12122.
57. A. Sumboja, C. Y. Foo, X. Wang and P. S. Lee, *Adv. Mater.*, 2013, **25**, 2809-2815.

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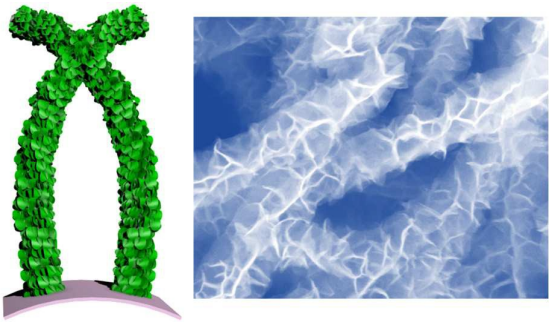

For the first time two phases MnO_2 nanostructures are constructed into core-branch arrays which deliver high pseudocapacitance.

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